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A Nanocrystalline Fe₂O₃ Film Anode Prepared by Pulsed Laser Deposition for Lithium-Ion Batteries

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Abstract

Nanocrystalline Fe₂O₃ thin films are deposited directly on the conduct substrates by pulsed laser deposition as anode materials for lithium-ion batteries. We demonstrate the well-designed Fe₂O₃ film electrodes are capable of excellent high-rate performance (510 mAh g⁻¹ at high current density of 15,000 mA g⁻¹) and superior cycling stability (905 mAh g⁻¹ at 100 mA g⁻¹ after 200 cycles), which are among the best reported state-of-the-art Fe₂O₃ anode materials. The outstanding lithium storage performances of the as-synthesized nanocrystalline Fe₂O₃ film are attributed to the advanced nanostructured architecture, which not only provides fast kinetics by the shortened lithium-ion diffusion lengths but also prolongs cycling life by preventing nanosized Fe₂O₃ particle agglomeration. The electrochemical performance results suggest that this novel Fe₂O₃ thin film is a promising anode material for all-solid-state thin film batteries.

Keywords: Lithium-ion batteries, Nanocrystalline Fe₂O₃, Anode material

Background

With the ever-increasing applications of lithium-ion batteries (LIBs) in portable electronics and electric vehicles, there has been extensive research on developing advanced electrode materials with higher energy and power densities [1–7]. Since the first report on reversible lithium storage in transition metal oxides (TMOs) by Poizot et al. [8], TMOs (Co₃O₄ [9, 10], NiO [11, 12], Fe₂O₃ [13–15], and CuO [16, 17]) have been widely explored as anode materials due to their higher theoretical specific capacity and better safety in comparison with traditional carbon anode materials. Among all these TMOs, Fe₂O₃ received much attention in recent years due to its high theoretical specific capacity (~ 1005 mAh g⁻¹), low cost, abundant resources, and environmental benignity. However, like other TMOs, the huge volume variations associated with Li-ion insertion/extraction often leads to the pulverization and subsequent falling off of the active materials from the electrode, which results in a significant capacity fade, poor cycling stability, and poor rate capability. To circumvent these

problems, many nanostructures of Fe₂O₃ have been synthesized for lithium-ion batteries, such as nanorods [18, 19], nanoflakes [20, 21], hollow sphere [22–24], core-shell arrays [25], and micro-flowers [26].

Besides all the above nanostructures, nanocrystalline thin film anodes (NiO [27], MnO [28], Cr₂O₃ [29], CoFe₂O₄ [30], Si [31], and Ni₂N [32]) deposited directly on conducting substrates by pulsed laser deposition or sputtering can also exhibit an excellent electrochemical performance due to the enhanced electrical contact between the substrates and active materials, the shortened diffusion lengths for lithium-ion, and the structure stability. What is more important is that thin films of TMOs have potential applications in all-solid-state microbatteries as self-supported electrodes [33, 34]. The TMOs' films can replace the lithium film anode which limits the integration of microbatteries with circuits due to the low melting point and strong reactivity with moisture and oxygen. However, up to now, there have been few reports on the Fe₂O₃ film anodes deposited by pulsed laser deposition or sputtering, and the reported specific capacities were much lower than the theoretical specific capacity of Fe₂O₃ [35, 36].

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In this work, we prepared nanocrystalline Fe_2O_3 films by pulsed laser deposition (PLD) as an anode material for lithium-ion batteries. The Fe_2O_3 thin film anodes with average grain size of several tens of nanometers showed high reversible capacity of 905 mAh g^{-1} at 100 mA g^{-1} and high rate capacity of 510 mAh g^{-1} at 15000 mA g^{-1} . The remarkable electrochemical performance demonstrates that nanocrystalline Fe_2O_3 thin film has potential applications in high performance LIBs, especially all-solid-state thin film batteries.

Experimental

Synthesis of Nanocrystalline Fe_2O_3 Films

The films of Fe_2O_3 were deposited directly on copper foils or stainless steels by a PLD technique in oxygen ambient. A KrF excimer laser with a wavelength of 248 nm was focused on the rotatable target of metal Fe. The repetition rate was 5 Hz, and the laser energy was 500 mJ. The distance between the target and the substrate was 40 mm. In order to get nanocrystalline Fe_2O_3 films, we grew samples at room temperature under oxygen pressure of 0.3 Pa on both copper foil and stainless steels. They showed the same electrochemical performance. The thickness of the nanocomposite film is approximately 200 nm as determined by atomic force microscope (AFM, Park systems XE7). The mass of 0.121 mg was obtained by measuring the difference of substrate before and after deposition via electrobalance (METTLER TOLEDO).

Material Characterization

The crystalline phase of the Fe_2O_3 film was characterized by X-ray diffraction (XRD) on a Rigaku D/Max diffractometer with filtered Cu $K\alpha$ radiation ($\lambda = 1.5406 \text{ \AA}$) at a voltage of 40 kV and a current of 40 mA. High-resolution transmission electron microscopy (TEM) and selected area electron diffraction (SEAD) were carried out by a JEOL 100CX instrument. For the TEM measurement, the Fe_2O_3 film grown on NaCl substrate was put into water to dissolve the NaCl. After that, the suspension was dropped onto a holey carbon grid and dried. The morphology of the samples were observed by scanning electron microscopy (SEM) using a SU8010. X-ray photoelectron spectroscopy (XPS) measurement was performed on a Thermo Scientific ESCALAB 250XI photoelectron spectrometer.

Electrochemical Measurements

For the electrochemical measurements, conventional CR2032 type coin cells with the Fe_2O_3 nanocrystalline film anodes were assembled inside an argon-filled glove box with the oxygen and moisture content below 0.1 ppm. The electrochemical cells were prepared using lithium metal as the counter electrode and a standard electrolyte of 1:1:1 ethylene carbonate (EC)/dimethyl carbonate (DMC)/ LiPF_6 . Galvanostatic cycling measurements were

processed at room temperature by a LAND-CT2001A battery system at various current rates between 0.01 and 3.0 V. Cyclic voltammetry (CV) and AC impedance measurements were performed with a CHI660E electrochemical workstation (CHI Instrument TN). The scanning rate was 0.1 mV s^{-1} .

Results and Discussion

X-ray diffraction (XRD) patterns of the Fe_2O_3 film are shown in Fig. 1a. It can be observed that there is no obvious peak except the peaks of cubic crystal Cu substrate, suggesting that the Fe_2O_3 film is amorphous or crystallized with nanosized grains. Such phenomenon could be attributed to the deposition occurred at room temperature. In order to determine the chemical composition of the obtained film, XPS measurement was performed as shown in Fig. 1b. The Fe $2p_{3/2}$ and Fe $2p_{1/2}$ main peaks are clearly accompanied by satellite structures on their high binding-energy side, with a relative shift of about 8 eV. The peaks of Fe $2p_{3/2}$ locating at 710.9 eV and Fe $2p_{1/2}$ locating at 724.5 eV are similar with XPS spectra of Fe_2O_3 reported in

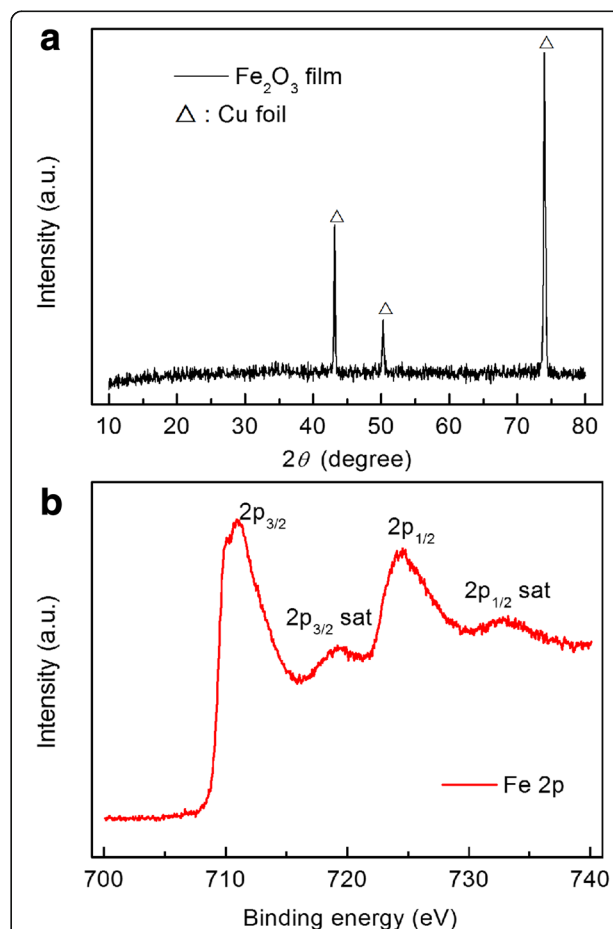


Fig. 1 Structure and composition characterization of Fe_2O_3 film deposited at room temperature. **a** XRD patterns of Fe_2O_3 film. **b** XPS spectrum of Fe_2O_3 film

the literature [37–39]. To further reveal the structure and composition of as-deposited thin films, TEM characterization was conducted as shown in Fig. 2. It revealed that the Fe_2O_3 films were made of small nanograins with average size of several tens of nanometers. The HRTEM image clearly presents the lattice fringes of the (110) corresponding to d-spacing of 0.251 nm of $\alpha\text{-Fe}_2\text{O}_3$. Meanwhile, the ring-like feature of the selected area electron diffraction (SAED) confirmed the polycrystalline nature of Fe_2O_3 film. As shown by the SEM images in Fig. 2c, the Fe_2O_3 film consists of particles in nanometer scale. Based on all these results, we can confirm that the film deposited at room temperature is composed of Fe_2O_3 with ultrafine nanosized crystalline grains.

The electrochemical performance of the electrode made of Fe_2O_3 nanocrystalline film was firstly evaluated by cyclic voltammetry (CV). Figure 3 shows the first three CV curves of Fe_2O_3 nanocrystalline film anode. The CV curves are similar to the previous reports of Fe_2O_3 anode [40–46]. In the first cathodic process, three peaks were observed at 1.38, 1.02, and 0.84 V, which could be related to a multi-step reaction. First, the very small peak at 1.38 V may be due to the lithium insertion into the crystal structure of Fe_2O_3 film forming $\text{Li}_x\text{Fe}_2\text{O}_3$ without change in the structure [40, 43]. Second, another peak at about 1.02 V could be ascribed to phase transition from hexagonal $\text{Li}_x\text{Fe}_2\text{O}_3$ to cubic LiFe_2O_3 . The third sharp reduction peak at 0.84 V corresponds to the complete reduction of iron from Fe^{2+} to Fe^0 and the formation of solid electrolyte interface (SEI). In the anodic process, two broad peaks observed at 1.57 and 1.85 V represent the oxidation of Fe^0 to Fe^{2+} and further oxidation to Fe^{3+} . In the subsequent cycles, the reduction peaks were replaced by two peaks locating around 0.88 V because of the irreversible phase transformation in the first cycle. The overlapping of the CV curves during the following 2 cycles demonstrated good reversibility of the electrochemical reactions, and this was further confirmed by the cycling performance.

Figure 4a shows the discharge and charge profiles of the Fe_2O_3 nanocrystalline film for different cycles at a specific current of 100 mA g^{-1} with a voltage range of 0.01–3 V. Obvious voltage hysteresis are observed due to the conversion reaction during charge/discharge processes, and the voltage plateaus are in good agreement with the above CV results. The clear voltage slopes observed in each charge/discharge process indicate the oxidation of Fe to Fe^{3+} and the reduction of Fe^{3+} to Fe, respectively. The smooth slope from 1.5 to 2.0 V in the charge process represents the two oxidation peaks in the CV curves. Meanwhile, the plateau or slope around 0.9 V in the discharge process represents the reduction peak in the CV curves. The initial discharge and discharge capacity of the Fe_2O_3 nanocrystalline film are 1183 and 840 mAh g^{-1} ,

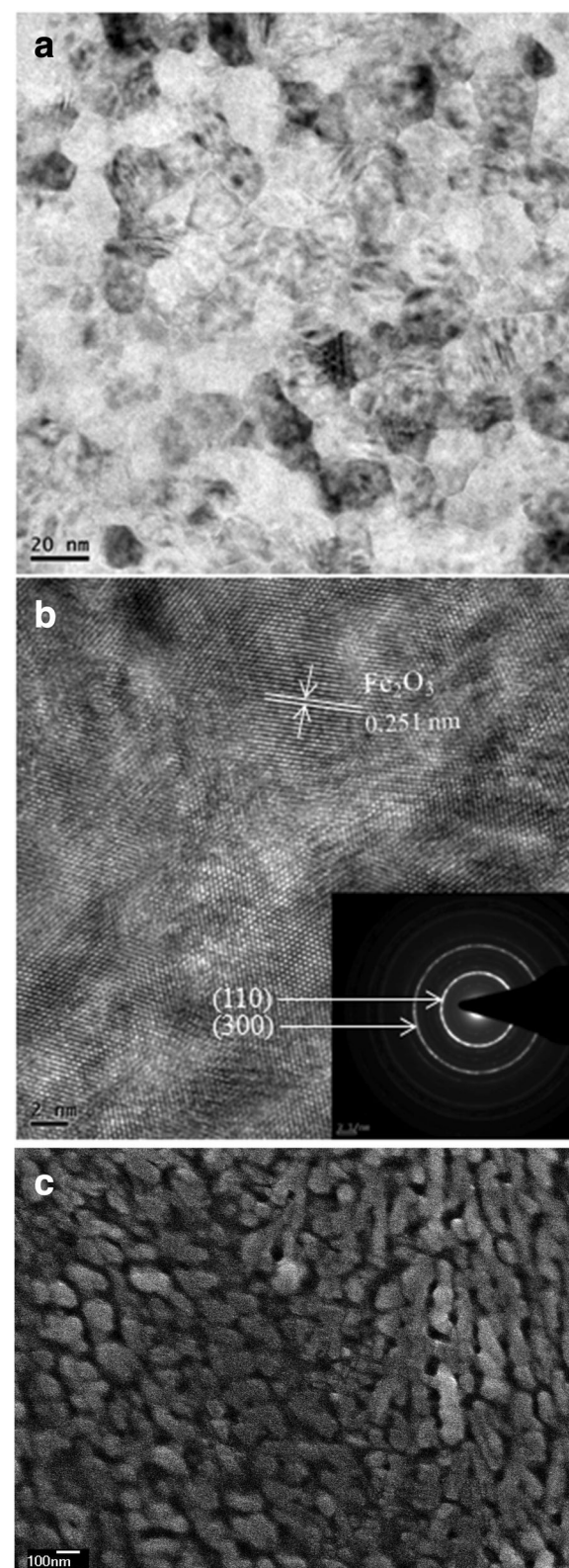
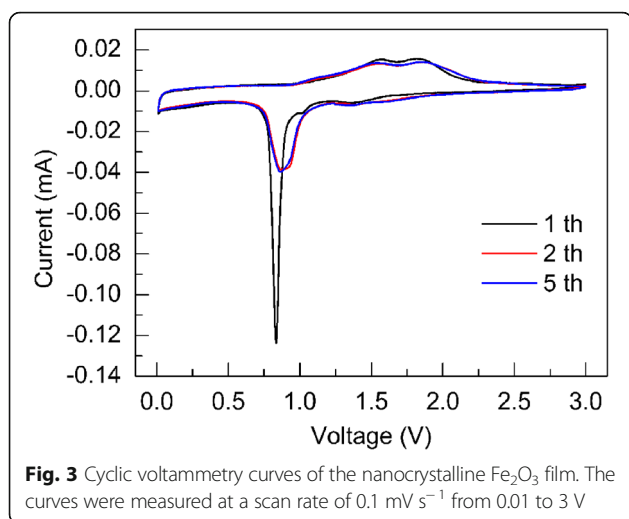


Fig. 2 a TEM image. b HRTEM image with inset showing SAED patterns. c SEM image of the Fe_2O_3 film prepared at room temperature



respectively, resulting in a Coulombic efficiency of 71%. The irreversible capacity loss is mainly attributed to the formation of SEI layer on the surface of anode, which is commonly observed in most anode materials [44–47].

The cycling performance of the film electrode at a specific current of 100 mA g^{-1} at room temperature is shown in Fig. 4b. It can be seen that the reversible capacity

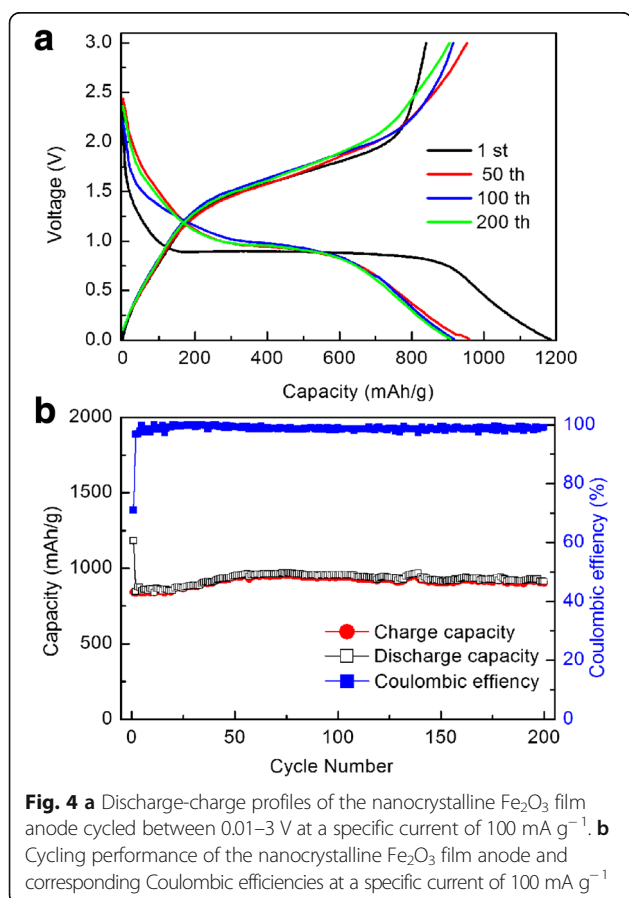
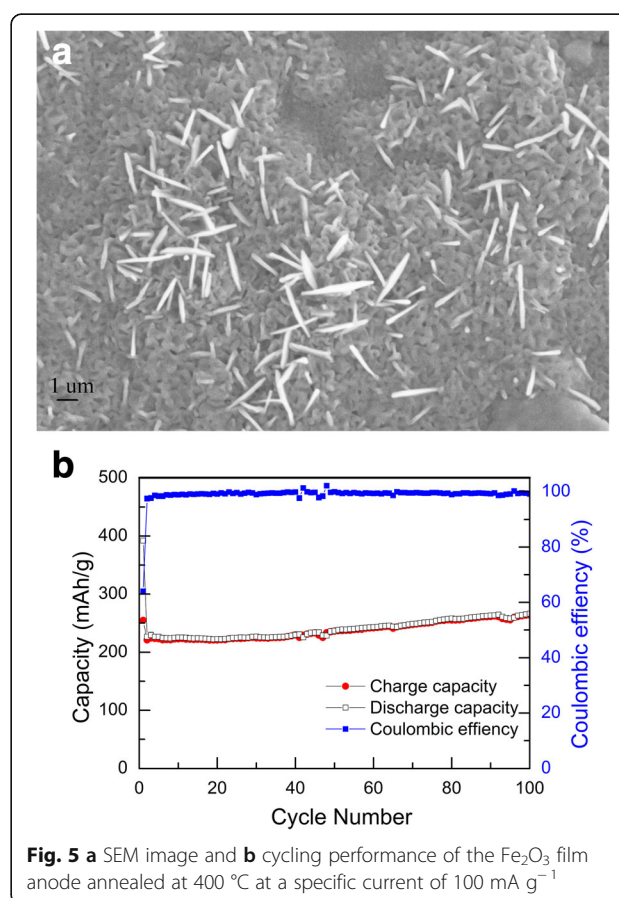


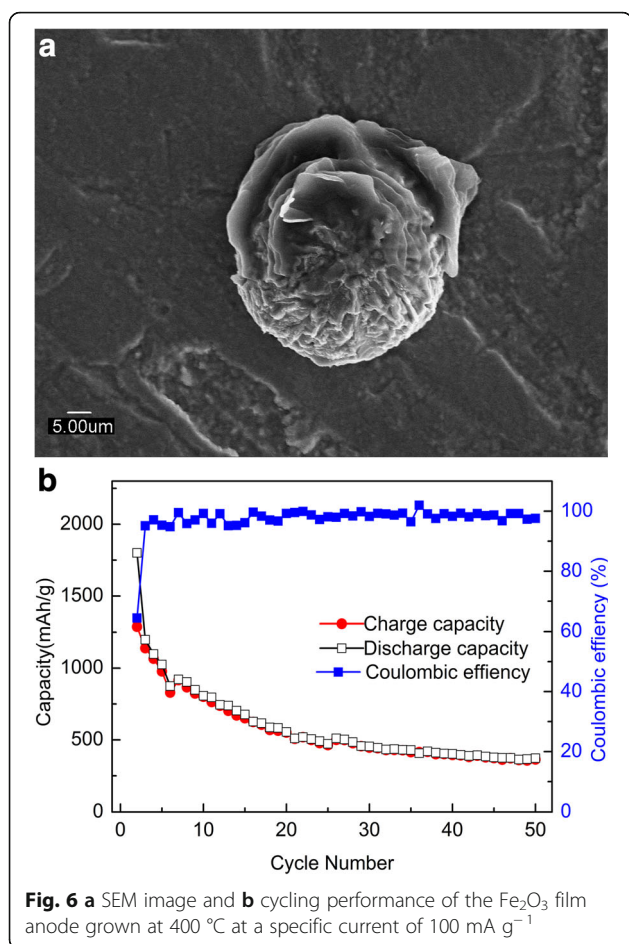
Table 1 The capacity comparison of our work with reported Fe_2O_3 film anode

Fe_2O_3 -based thin film anode	Current density	Cycle number	Capacity (mAh g^{-1})	Ref.
Pulsed laser deposition	100 mA g^{-1}	200	905	This work
Pulsed laser deposition	100 mA g^{-1}	50	361	
Pulsed laser deposition	$100 \mu\text{A cm}^{-2}$	50	280	[35]
Sputter deposition	165 mA g^{-1}	120	330	[36]

gradually increases to 951 mAh g^{-1} after the 70 cycles and then keeps stable in the range of $900\text{--}950 \text{ mAh g}^{-1}$ with a Coulombic efficiency nearly 100% during the following cycles. Similar phenomenon of the capacity increasing during cycling has been found in many transition metal oxide electrodes in previous studies [13, 48–52]. The possible reason for this would be the electrode activation, which induces the reversible growth of polymer/gel-like films to increase the capacity at low potentials [50]. Compared with the previous reports of Fe_2O_3 film anode batteries deposited by pulsed laser deposition or sputtering [35, 36], the capacity of Fe_2O_3 in our work has a considerable improvement as summarized in Table 1.

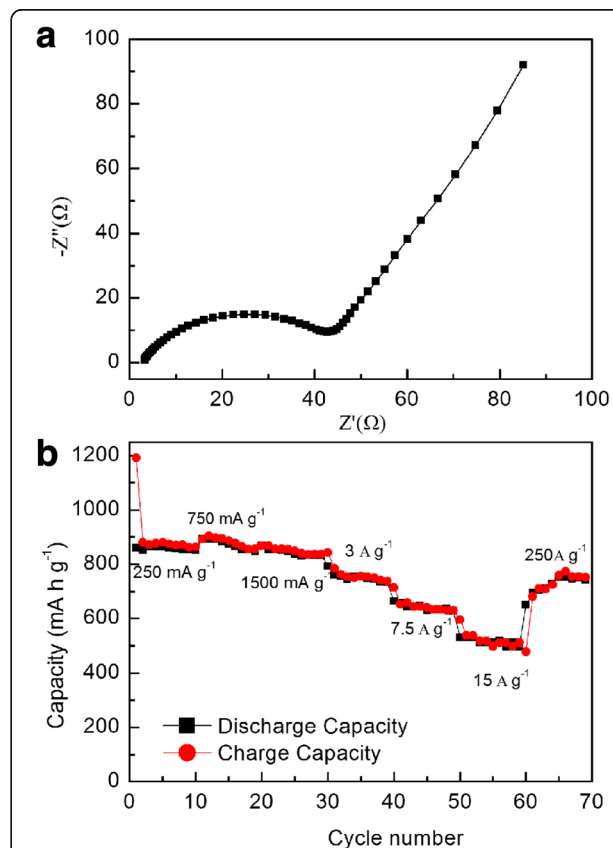
Previous studies on the effect of particle size on lithium intercalation into Fe_2O_3 shows that nanocrystalline Fe_2O_3 exhibited better electrochemical performance





than macro-sized ($> 100\text{ nm}$) Fe_2O_3 [53]. To confirm the role of particle size in the electrochemical performance, we annealed the as-prepared Fe_2O_3 film on stainless steels at 400° . The prepared Fe_2O_3 film anode at high temperature was deposited on stainless steels only due to the instability of copper foil. The morphology comparison in Fig. 5a and Fig. 2c confirms that the particle sizes of the samples annealed at high temperature are obviously larger. Figure 5b shows that the capacities was only about 263 mAh g^{-1} after 100 circles, which was much lower than the specific capacity of as-prepared Fe_2O_3 . In addition, we also fabricated Fe_2O_3 film anode with larger particle size on stainless steels under 400°C as shown in Fig. 6a. Figure 6b shows its discharge and charge profiles for different cycles at a specific current of 100 mA g^{-1} . The capacities dropped to 361 mAh g^{-1} after 50 circles. These results indicate that the enhanced reversible capacity of nanocrystalline Fe_2O_3 film grown at room temperature can be attributed to the nanoscaled structure of the thin film electrode, which can sustain high lithium insertion strain because of the smaller number of atoms and large surface areas within nanoparticles [13, 14, 54].

To investigate the kinetics of lithium inserting/deinserting, electrochemical impedance spectra measurement was performed in Fig. 7a. The charge-transfer impedance on the electrode/electrolyte surface is about $50\ \Omega$, which can be deduced from the single semicircle in the high-middle frequency. The superior conductivity of the film electrode without binder can be attributed to the nanocrystalline structure of the Fe_2O_3 film and the enhanced electrical contact between active anode and substrate. The good conductivity of the nanocrystalline Fe_2O_3 film anode led to excellent rate performance. Figure 7b shows the charge/discharge capacities at different current densities. The anode delivered capacities up to 855, 843, 753, 646, and 510 mAh g^{-1} at high current densities of 750, 1500, 3000, 7500, and $15,000\text{ mA g}^{-1}$, respectively, which is corresponding to 98.2, 96.7, 87.8, 75.3, and 59.5% retention of the capacity at 250 mA g^{-1} (about 871 mAh g^{-1}). More importantly, when the specific current reduced to 250 mA g^{-1} , the capacity could recover to 753 mAh g^{-1} . The excellent rate performance benefits from both the good conductivity of the anode and the increase of capacity upon cycling.



Conclusions

In summary, nanocrystalline Fe_2O_3 film anode has been deposited by pulsed laser deposition at room temperature. The results of structure and morphology characterization showed that the deposited films are composed of nanocrystalline Fe_2O_3 with grain size of several tens of nanometers. The prepared Fe_2O_3 exhibits an excellent electrochemical performance, such as superior cycling stability (905 mAh g^{-1} at a specific current of 100 mA g^{-1} after 200 cycles) and high rate capability (510 mAh g^{-1} at 15000 mA g^{-1}). The outstanding electrochemical performance can be related to the nanocrystalline structure of Fe_2O_3 which could sustain high strain, shorten diffusion lengths for lithium-ion, and keep the structure stable. The excellent electrochemical performance and room temperature growth suggest that nanocrystalline Fe_2O_3 has potential application in high performance LIBs, especially in all-solid-state thin film batteries.

Abbreviations

AFM: Atomic force microscope; CV: Cyclic voltammetry; DMC: Dimethyl carbonate; EC: Ethylene carbonate; LIB: Lithium-ion batteries; PLD: Pulsed laser deposition; SEAD: Selected area electron diffraction; SEI: Solid electrolyte interface; SEM: Scanning electron microscopy; TEM: Transmission electron microscopy; TMOs: Transition metal oxides; XPS: X-ray photoelectron spectroscopy; XRD: X-ray diffraction

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Availability of Data and Materials

All data are fully available without restriction.

Authors' Contributions

The experiments and characterization presented in this work were carried out by XLT, YZQ, XTS, and STF. The experiments were designed by QL and SDL. The experiments were discussed in the results by XW, HSL, JX, and DRC. All authors read and approved the final manuscript.

Authors' Information

Not applicable

Competing Interests

The authors declare that they have no competing interests.

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